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layers in single-crystal diamond

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2 **Effects of high-power laser irradiation on sub-superficial graphitic layers in single-crystal**

3 **diamond**

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5 F. Picollo[1,2,3*], S. Rubanov[4], C. Tomba[5,6,7], A. Battiato[2,1,3], E. Enrico[8], A. Perrat-Mabilon[9,10],

6 C. Peaucelle[9,10], T. N. Tran Thi [5,9,11], L. Boarino[8], E. Gheeraert[5,7], P. Olivero[2,1,3,8]

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9 *[1] National Institute of Nuclear Physics (INFN), section of Torino, Italy*

10 *[2] Physics Department and "NIS" Inter-departmental centre, University of Torino, Torino, Italy*

11 *[3] National Interuniversity Consortium for the Physical Sciences of Matter (CNISM), Torino Unit, Italy*

12 *[4] Bio21 Institute, University of Melbourne, Australia*

13 *[5] University of Grenoble Alpes, F-38000 Grenoble, France*

14 *[6] Laboratoire des Technologies de la Microelectronique, Minatec Campus, F-38054 Grenoble, France*

15 *[7] CNRS, Institute NEEL, F-38042 Grenoble, France*

16 *[8] National Institute of Metrologic Research (INRiM), Torino, Italy*

17 *[9] University of Lyon 1, CNRS, Inst Phys Nucl Lyon, F-69622 Villeurbanne, France*

18 *[10] IN2P3, F-69622 Villeurbanne, France*

19 *[11] European Synchrotron Radiation Facility (ESRF), Grenoble, France*

20 **Abstract**

21 We report on the structural modifications induced by a $\lambda = 532$ nm ns-pulsed high-power laser

22 on sub-superficial graphitic layers in single-crystal diamond realized by means of MeV ion

23 implantation. A systematic characterization of the structures obtained under different laser

24 irradiation conditions (power density, number of pulses) and subsequent thermal annealing was

25 performed by different electron microscopy techniques. The main feature observed after laser

26   irradiation is the thickening of the pre-existing graphitic layer. Cross-sectional SEM imaging was

27   performed to directly measure the thickness of the modified layers, and subsequent selective

28   etching of the buried layers was employed to both assess their graphitic nature and enhance the

29   SEM imaging contrast. In particular, it was found that for optimal irradiation parameters the laser

30   processing induces a six-fold increase the thickness of sub-superficial graphitic layers without

31   inducing mechanical failures in the surrounding crystal. TEM microscopy and EELS

32   spectroscopy allowed a detailed analysis of the internal structure of the laser-irradiated layers,

33   highlighting the presence of different nano-graphitic and amorphous layers. The obtained results

34   demonstrate the effectiveness and versatility of high-power laser irradiation for an accurate

35   tuning of the geometrical and structural features of graphitic structures embedded in

36   single-crystal diamond, and open new opportunities in diamond fabrication.

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40   * corresponding author: picollo@to.infn.it

41   ph: +39 011 670 7879

42   fax: +39 011 670 7020

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1. Introduction

Diamond is well known for its range of extreme mechanical, thermal and optical properties, which make it an attractive material for a variety of applications [1]. Nevertheless, diamond is a metastable allotropic form of carbon at standard pressure and temperature, and can be converted into graphite if an energy barrier is overcome [2]. Several approaches have been developed to induce this phase transition, among which ion-beam-induced graphitization [3–9] and laser-induced graphitization [10,11] play a prominent role. The former approach takes advantage of the ion-induced defect creation caused by nuclear collisions to amorphize the material and the subsequent thermal annealing to convert amorphized regions into a graphitic phase [3]. The latter approach is based on complex non-equilibrium dynamics induced by high-power light absorption, which were modelled with different theoretical approaches based on the non-radiative recombination of electron-hole pairs [11] or on a non-thermal ultrafast non-equilibrium phase transition [12,13].

Several previous studies explored the laser-induced graphitization process of single-crystal diamond: the first investigations dating back to the 80's were focused on realization of graphitic structures on diamond surface with direct writing or optical projection by means of excimer lasers [14], approached that was further investigated also in recent years [15]. Subsequently, new theoretical models of pulsed laser irradiation were proposed taking into account fast energy transfer mechanisms [16]. Consequently, in the last decade several works were carried out to exploit the possibility of realizing three-dimensional structures into diamond bulk by means of femtosecond [17,18] and picosecond [18–20] pulsed laser writing. Furthermore, the possibility of enhancing the resolution in the laser fabrication of the graphitic structures with the use of adaptive optical elements was recently demonstrated [21,22].

73   Direct laser-induced graphitization represents an extremely versatile technique with promising

74   applications in different fields such as the realization of diamond-based particle detectors [23–

75   27] and (upon the selective removal of the graphite) microfluidics devices for biomedical sensing

76   [28].

77   On the other hand, this technique is limited by the poor geometrical quality of structures

78   finishing, which is inherently caused by the nature of the graphitization process [18,20]. In order

79   to overcome this limitation, laser-induced graphitization in diamond can be combined with a

80   preliminary MeV-ion-induced graphitization stage. By taking advantage of the high degree of

81   control on the geometrical properties (depth, thickness) of MeV-ion-induced buried graphitic

82   structures in diamond allowed by the peculiar nuclear energy loss profile of MeV ions [29], this

83   double-step procedure guarantees a better definition in the material micro-structuring [30] and

84   also represents an interesting improvement in the realization of particle detectors [26,31,32],

85   bolometers [33,34], bio-sensors [35–37], metallic-dielectric structures [38] and microfluidics

86   [39].

87   In the present paper we report on the use of ns-pulsed laser irradiation for the structural

88   modification and thickening of sub-superficial graphitic layers in diamond, which were realized

89   by means of MeV ion implantation. The above-mentioned structures are imaged before and after

90   the selective removal of the graphitic phase with respect to the surrounding diamond matrix, and

91   are characterized in their structural properties by transmission electron microscopy.

92

93   **2. Experimental**

94   In the present study, a commercial synthetic (001) single-crystal diamond grown by High

95   Pressure High Temperature method (HPHT) by ElementSix (Ascot, UK) was used. The diamond

96   is $3\times3\times0.3$ mm^3 in size and it is classified as type Ib, having a nominal substitutional nitrogen

97   concentration between 10 ppm and 100 ppm. The sample is cut along the 100 crystal direction

98   and it is optically polished on the two opposite large faces.

99   The sample was implanted at room temperature at the "Service Faisceaux d'Ions" laboratory of

100   the Nuclear Physics Institute (University of Claude Bernard Lyon 1) across one of the two main

101   polished surfaces with a broad 2 MeV He$^+$ ion beam to deliver a uniform fluence of 1×10^{17} cm^{-2}

102   across the irradiated area. During the implantation, the beam current was ~200 Na. The process

103   of damage induced by MeV ions in matter occurs mainly at the end of ion range, where the cross

104   section for nuclear collisions is strongly enhanced, after the ion energy is progressively reduced

105   by electronic interactions occurring in the initial stages of the ion path [40]. Figure 1 shows the

106   strongly non-uniform depth profile of the density of induced vacancies ($\#_{vac}$ cm^{-3}) evaluated in a

107   linear approximation as the product between the implantation fluence ($\#_{ions}$ cm^{-2}) and the linear

108   density of induced vacancies per single ion ($\#_{vac}$ cm^{-1} $\#_{ions}^{-1}$). The latter quantity was estimated

109   with the "Stopping and Range of Ions in Matter" (SRIM)-2013.00 Monte Carlo code [29] in

110   "Detailed calculation with full damage cascade" mode by taking an atom displacement energy

111   value of 50 Ev [41]. The high density of damage induced by ion implantation promotes the

112   conversion of the diamond lattice to an amorphous phase, which is located ~3.5 μm below the
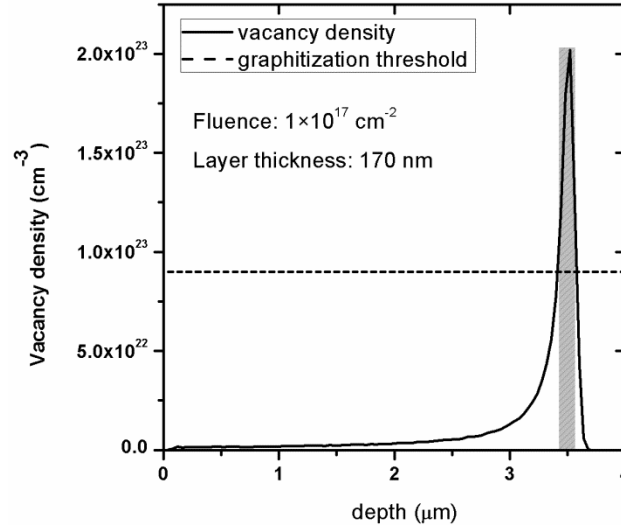
113   sample surface.

114

Fig. 1: Depth profile of the volumetric vacancy density induced in diamond by 2 MeV He[+] implanted at a fluence of

1×10^{17} cm^{-2}. The graphitization threshold is reported in dashed line. The amorphized region is highlighted by the

grey area in correspondence of the intersection of the Bragg peak with the graphitization threshold.

118

119 The above-mentioned implantation fluence allowed to overcome a critical damage density,

120 usually referred as "graphitization threshold" [42], whose value in the above-mentioned linear

121 approximation has been estimated as $\sim 9 \times 10^{22}$ cm^{-3} for light MeV irradiation [43], as indicated in

122 Figure 1. Such a model of the damage profile has to be considered as a rough estimation since it

123 results from a linearly cumulative effect of ion damage, i.e. by neglecting any damage saturation

124 effects occurring at high damage levels such as self-annealing and vacancies interactions [44,45].

125 Nonetheless, in this context it can be considered as a satisfactory approach to estimate the depth

126 and thickness of the buried region. After ion implantation, the sample was thermally annealed for

127 1 hour at a temperature of 900 °C, which is suitable for the conversion of amorphous carbon to a

128 graphitic phase, as confirmed by TEM studies [46–48]. Concurrently, the annealing process

129 restores the pristine diamond structure in the lightly-damaged cap layer, i.e. the region comprised

130   between the surface and the buried graphitic layer [49,50]. The process was carried out in

131   vacuum (p ≈ 10^{-6} mbar) to avoid accidental etching of the diamond surface due to oxidation.

132   The ion-implanted side of the sample was subsequently irradiated with nanosecond-pulsed

133   Nd:YAG laser (EzLaze3 by New Wave) equipped with a Q-switching system. This laser source

134   generates pulses of 4 ns duration with a repetition rate of either 1 Hz or 5 Hz. Two different

135   emission wavelengths can be selected, i.e. 1064 nm and 532 nm, the second one being obtained

136   by means of an angle-tuned KTP crystal. The laser beam is focused onto the sample with a

137   microscope supplied with 5×, 20× and 100× objective lenses. The co-axial imaging through the

138   microscope offers the opportunity of monitoring the sample processing in real time. Using the

139   532 nm wavelength and the 100× objective, the maximum emission power is 150 Kw and the

140   minimum size of the spot is 5×5 μm^2, thus resulting in a maximum power density of

141   ~22 GW cm^{-2}.

142   A Quanta 3D™ dual-beam system by FEI available at the "NanoFacility Piemonte" laboratories

143   of the INRiM Institute was employed to cross-section the sample by 30 keV Ga^+

144   focused-ion-beam (FIB) milling and to estimate the thickness of the graphitic layer before and

145   after the laser irradiation by SEM imaging .

146   The selective electrochemical etching of the graphite was performed with the purpose of

147   enhancing the topographical contrast in the SEM imaging of the buried layers. This process was

148   performed by immerging the sample in a water solution of H_3BO_3 (4×10^{-3} mol l^{-1} concentration)

149   for 1 hour applying a DC voltage of 150–200 V through a couple of platinum electrodes placed

150   in close proximity of the sample [51].

151   The TEM imaging was performed at the Microscopy laboratories of the Bio21 Institute

152   (University of Melbourne) for a detailed study of the thickness and the structure of the graphitic

153 layers before and after laser annealing. To this scope, a Tecnai TF20 electron microscope

154 operated at 200 keV was employed. Cross-sectional TEM samples with thickness ~100 nm were

155 prepared in [110] and [100] orientations using a standard FIB lift-off technique. Selected area

156 diffraction (SAD) patterns were collected with smallest aperture (diameter ~180 nm in the

157 specimen plane). Nano-beam diffraction patterns were collected in nano-beam scanning TEM

158 (STEM) mode with beam size ~10 nm. Electron energy loss spectroscopy (EELS) was conducted

159 by employing a Gatan Enfina energy filter.

160

161 **3. Results and discussion**

162 FIB milling was employed to allow the cross-sectional SEM imaging of the ion-implanted and

163 subsequently annealed regions, as shown in Fig. 2a. The presence of the graphitic layer

164 embedded in the diamond matrix can be recognized in the dark grey horizontal strip, although

165 the low contrast in the secondary emission yield between graphite and diamond does not allow

166 performing an accurate measure of its thickness.

167 Selective etching of the graphite was therefore performed in order to improve imaging contrast

168 and facilitate the measurement of the thickness of the sub-superficial layer. Fig. 2b shows a SEM

169 micrograph of the same region of the sample reported in Fig. 2a, after selective chemical etching

170 of the graphite layer. The previously graphitized region now corresponds to a gap within the

171 material, thus allowing a better visibility of its thickness and depth.
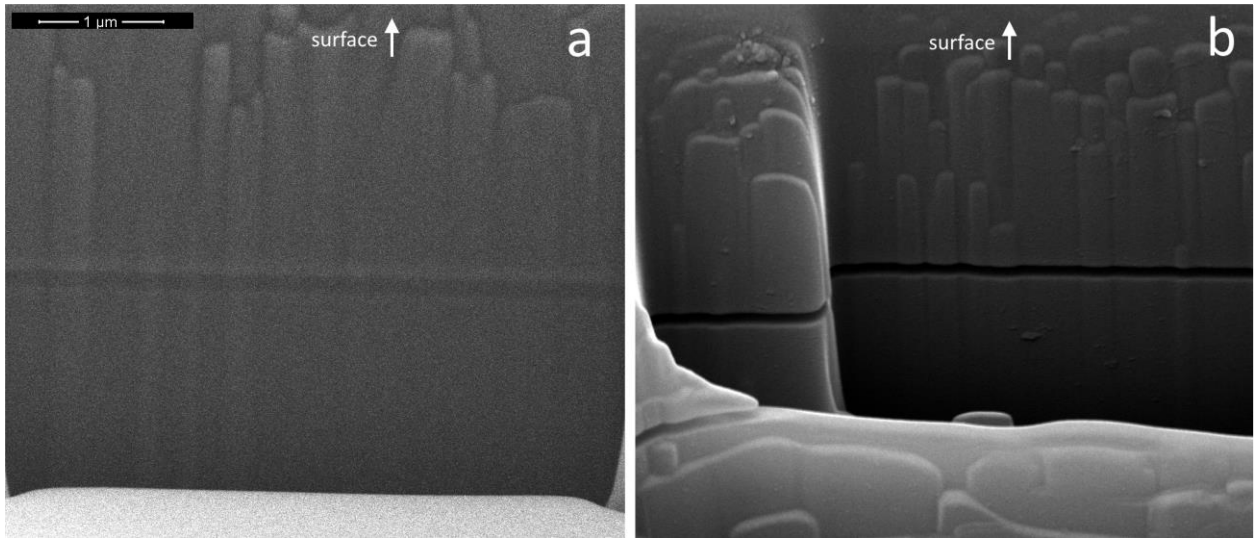
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Fig. 2: Cross-sectional SEM micrographs of the sub-superficial graphitic layer in ion-implanted and annealed diamond a) before and b) after the selective etching process. A better definition of the geometrical parameters of the buried structure is clearly obtained after its selective removal by means of the etching process.

The thickness of the buried layer evaluated from Fig. 2a is (200 ± 40) nm, while it was estimated (140 ± 20) nm from Fig. 2b. This discrepancy is due to the relaxation of the upper cap layer, as it will be more extensively discussed in the following.

A more detailed structural characterization of the investigated regions was performed by means of cross-sectional TEM microscopy and EELS spectroscopy. A bright-field TEM micrograph and the corresponding Selected Area Diffraction (SAD) pattern of the implanted layer after thermal annealing are shown in Figs. 3a and 3b, respectively. The implanted layer is clearly visible due to the higher contrast and its width was estimated as (130 ± 3) nm. As shown in Fig. 3b, the diffraction pattern is characterized by well-defined spots arising from the diffraction of the bulk diamond, as well as arcs along the {220} reflections from the graphitic C-planes of the implanted layer. As shown in Fig. 3c, the corresponding EELS spectrum from the buried graphitic layer of the same region has a prominent feature at ~285 Ev, which is characteristic of

190   sp^2 bonding. Therefore, TEM diffraction pattern combined with EELS confirm the conversion of

191   the implanted layer into a nanocrystalline graphitic phase with predominant orientation of

192   C-planes normal to the diamond surface.
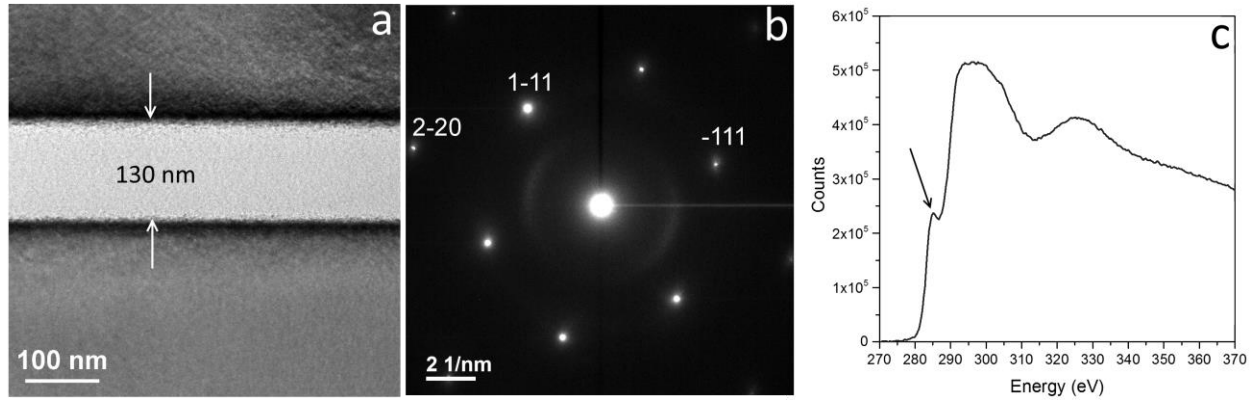
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194   Fig. 3: a) Bright-field TEM cross-section micrograph of the sub-superficial graphitic layer in diamond before laser

195   irradiation (the sample surface is located towards the top of the picture). The thickness of the ion-implanted and

196   subsequently annealed structure is highlighted, corresponding to (130 ± 3) nm. B) SAD pattern taken from the same

197   area, showing well defined spots and arcs respectively from the diamond and graphitic regions. C) EELS spectrum

198   of the carbon K-edge taken from the implanted layer; the characteristic feature at 285 eV is indicated by the arrow.

199

200   Multiple-pulse laser irradiation at $\lambda = 532$ nm wavelength of the ion-implanted and subsequently

201   annealed diamond sample was carried out at two different powers densities (namely,

202   0.41 GW cm^{-2} and 0.45 GW cm^{-2}) over an area of 26×26 μm^2.

203   Cross-sectional bright- and dark-field TEM micrographs of an area irradiated with 50 laser

204   pulses at a power density of 0.41 GW cm^{-2} are reported in Figs. 4a and 4b, respectively, and

205   show a complex multi-layer structure within the buried layer. The laser-induced graphitic layer,

206   formed in the region located directly above the ion-implanted layer, results in an overall

207   thickness of (690 ± 15) nm. At the bottom of Fig. 4b (i.e. towards the bulk of the sample), the

208   nanocrystalline graphite layer due to ion implantation is clearly distinguishable, exhibiting the

209 same thickness measured before the laser irradiation (see Fig. 2) and no appreciable differences
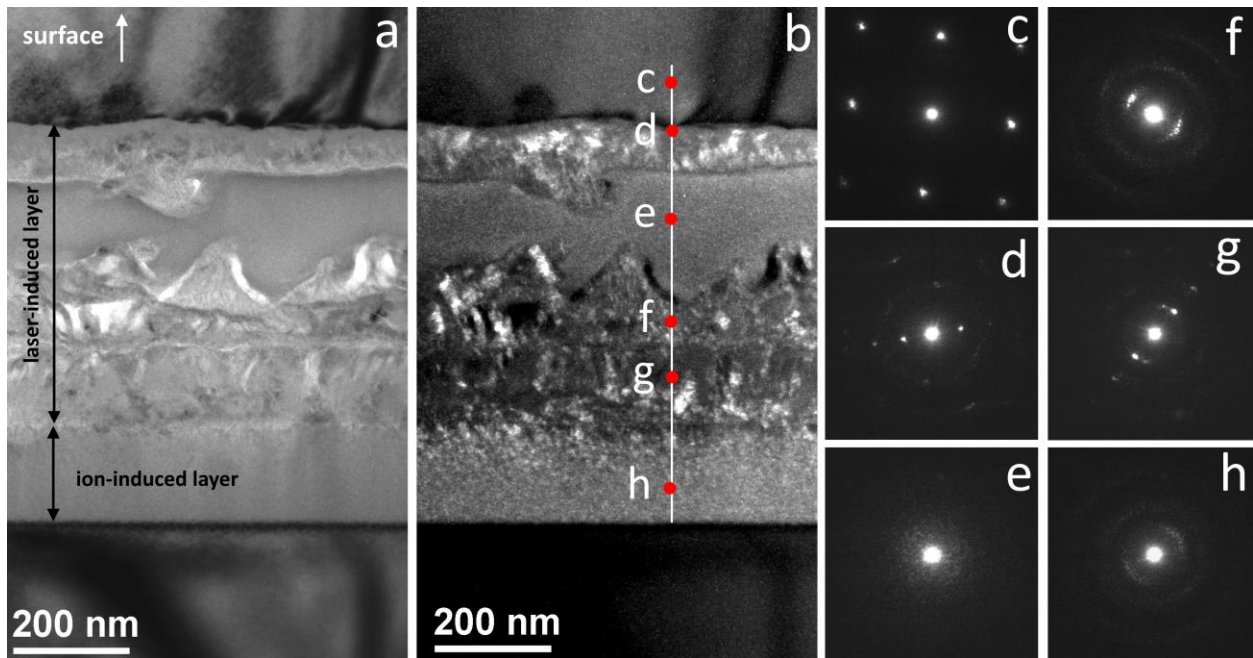
210 in its structure.

211

212 Fig. 4: a) Bright- and b) dark-field TEM cross section micrographs of a laser irradiated ($\lambda = 532$ nm, power density:

213 0.41 GW cm^{-2}, 50 pulses) nanocrystalline graphitic sub-superficial layer obtained after annealing at 900 °C of an

214 implanted diamond region. The layer thickens towards the sample surface (i.e. towards the top of the figure), with a

215 complex structuring of the modified region consisting in amorphous-carbon and nanocrystalline-graphite layers. The

216 numbered red dots in b) highlight the regions from which the diffraction patterns reported in c-h) were acquired.

217

218 The laser-induced phase transition is driven by thermal effects that originate as a consequence of

219 the strong absorption of the laser radiation by the implanted graphitic layer. Since the laser beam

220 irradiates the side of the nanocrystalline graphitic layer facing the sample surface, the induced

221 heating of the material is mainly localized within the "cap layer" comprised between the layer

222 itself and the sample surface rather than towards the bulk, and therefore graphitization mainly

223 occurs in such direction. An estimate of the temperature rise in the cap layer can be obtained

224 from the pressure-temperature phase diagram of elemental carbon [52], with a similar approach

225   to what is reported in [2]. Finite element simulation studies from diamond samples implanted

226   under the same experimental conditions [53] indicate that the regions surrounding the buried

227   graphitic layer experience pressures of 8–10 GPa due to its constrained volume expansion.

228   Furthermore, the experimental evidence indicates that the temperature rise is high enough to

229   drive the graphitization process within the corresponding stable portion of the phase diagram,

230   without incurring in its liquefaction. The laser-induced local temperature rise is therefore

231   estimated between 2500 °C and 4500 °C. Such a significant heating is expected to rapidly

232   dissipate due to the high thermal conductivity of diamond.

233   Nano-beam diffraction patterns were taken in scanning TEM mode along the line shown in

234   Fig. 4b, which is crossing perpendicularly the graphitic layer at 10 nm steps. A selection of the

235   obtained diffraction patterns are shown in Figs. 4c-h. The peculiar structure of the laser-induced

236   layer is characterized by the presence of an amorphous carbon phase (diffraction pattern in

237   Fig. 4e) comprised between two polycrystalline graphitic phases (diffraction patterns in Figs. 4d

238   and 4f), as clearly visible in the dark-field micrograph of Fig. 4b. The dark field micrograph was

239   constructed by selecting the graphitic arcs in the diffraction pattern, and therefore the graphitic

240   crystals appear as bright spots in it. These spots have <5 nm sizes in implanted layer and much

241   larger dimensions in the laser-induced layer. The observed multi-layer structure is somewhat

242   surprising and can be qualitatively attributed to a complex combination of thermal gradient and

243   stress effects occurring in the diamond during the pulsed laser irradiation [54].

244   A SEM micrograph of the same laser-irradiated region is reported in Fig. 5a, from which a layer

245   thickness of (700 ± 70) nm can be estimated, compatibly (although with lower accuracy) with

246   what obtained from TEM imaging. The amorphous areas are visible inside the graphitic layer, as

247   well as some voids. These voids were probably filled with re-deposited material during TEM

248   sample preparation and are therefore not visible in TEM images reported in Figs. 4a and 4b.

249   Following the same procedure adopted for the sample after implantation and annealing (see

250   Fig. 2), a selective electrochemical etching was carried out and the resulting structure is shown in

251   the TEM micrograph reported in Fig. 5b. The removal of the graphitic layer is evident in both

252   micrographs. Also, re-deposited material is visible on both surfaces of the gap, as commonly

253   observed in FIB milling. As shown in Fig. 5b, after laser irradiation and chemical etching the

254   buried layer was not entirely removed, but rather only the ion-induced nanocrystalline graphite

255   layer was etched, as confirmed by the comparison of the thicknesses of the regions reported in

256   Figs. 4a and 5b. This is a surprising result, since the polycrystalline graphitic phase formed

257   during laser irradiation is expected to be effectively etched by the electrochemical attack.

258   A thermal treatment at 900 °C for 2 hours was therefore performed on the same sample after

259   laser irradiation, with the purpose of inducing the graphitization of residual amorphous/distorted

260   material in the un-etched region. Afterwards, the sample was exposed to the same etching

261   process. As shown in Fig. 5c, the width of the etched layer in the laser-irradiated and annealed

262   sample increased up to (420 ± 20) nm, indicating a full removal of the laser-irradiated layer. As

263   already reported for the sample before laser irradiation, a discrepancy between the thickness

264   values of the laser-induced layer and the remaining gap is observed (see Figs. 5a and 5c). As

265   mentioned before, this effect can be explained by considering that a relaxation of the diamond

266   cap layer takes place after the removal of the graphitic layer. Since the graphite is characterized

267   by a significantly lower atomic density with respect to diamond, a volume expansion takes place

268   upon the graphitization process [55,56], thus deforming the upper diamond cap layer comprised

269   between the graphic layer and the sample surface. Once the graphite is removed, the cap layer

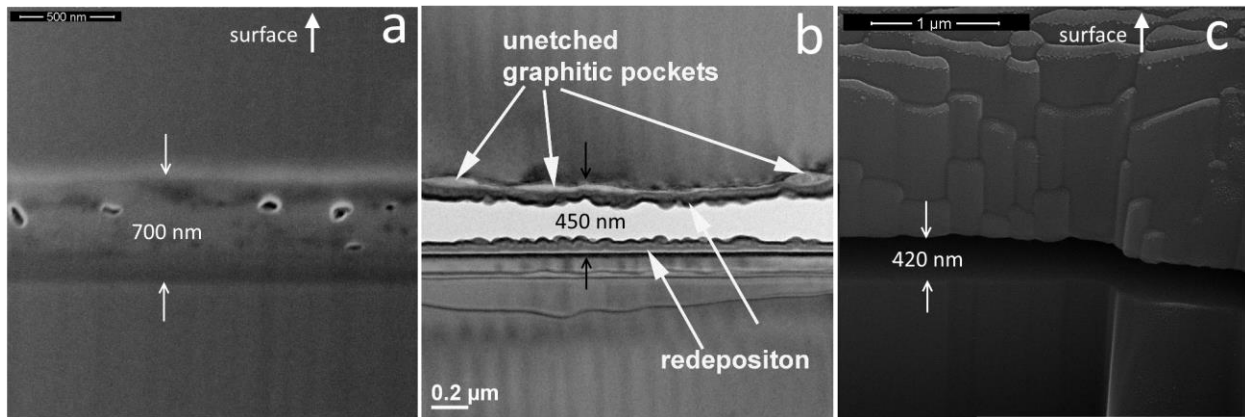270   undergoes a structural relaxation which induces the thinning of the etched region.

271

272   Fig. 5: a) Cross sectional SEM micrograph of the buried damaged layer after laser irradiation ($\lambda = 532$ nm, power

273   density: 0.41 GW cm^{-2}, 50 pulses). b) Cross-sectional TEM micrograph of the same layer reported in a) after

274   electrochemical etching; together with the characteristic re-deposited material present in the gap, un-etched

275   laser-induced layers are recognizable around the gap. c) Cross-sectional SEM micrograph of the same layer reported

276   in a) after a further 900 °C annealing step and electrochemical etching; no un-etched layers are visible.

277

278   Even more surprisingly, after the first etching process (i.e. before the second annealing step)

279   highly oriented graphitic clusters were found in correspondence of the partially un-etched

280   diamond/graphite interface which was closest to the sample surface, as shown in Fig. 5b.

281   Fig. 6a reports a high-resolution TEM micrograph of one of these clusters. The presence of

282   graphite C-planes is clearly visible in the corresponding diffraction pattern reported in Fig. 6c,

283   indicating the highly oriented structure of these graphitic clusters.
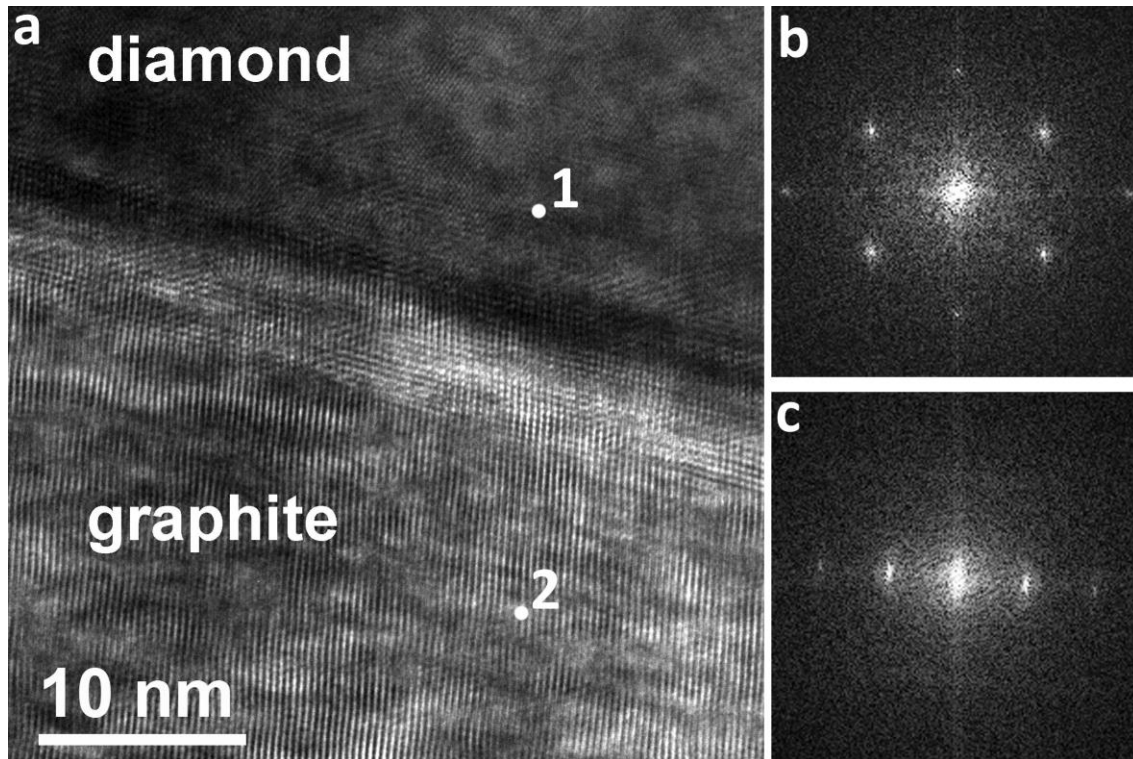
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Fig. 6. a) High resolution TEM micrograph of one of the highly oriented graphitic clusters remaining near the diamond interface after the chemical etching of the laser-irradiated sample. The corresponding diffraction patterns from diamond (spot "1") and graphite (spot "2") are reported in b) and c), respectively.

The results reported so far refer to regions of the sample which were exposed to 50 laser pulses and subsequently subjected to different annealing and/or etching processes. A systematic investigation of the laser-induced graphitization was also carried out as a function of the laser irradiation parameters (power, number of pulses). As previously reported, the implanted and subsequently annealed sample was irradiated with 532 nm laser pulses and power densities of 0.41 GW cm^{-2} and 0.45 GW cm^{-2} over different 26$\times$26 μm^2 areas. Following these laser irradiations, the sample was annealed at 900 °C for 2 hours with the purpose of inducing a full graphitization in the irradiated areas, consistently with the above-mentioned results. Also in this case, cross-sectional SEM imaging of the irradiated regions was systematically performed to

299   evaluate the thickening in regions processed with different laser irradiation conditions, and the

300   layer thicknesses were measured before and after the etching process.

301   Fig. 7 summarizes the results obtained for the two different power densities for a number of laser

302   pulses ranging between 1 and 300. It is evident that the thickness of the graphitic sub-superficial

303   layer obtained after laser irradiation and thermal annealing increases at increasing numbers of

304   laser pulses, up to more than 6 times its initial value. The trends reported in Fig. 7 clearly

305   indicate a strong sub-linear dependence of the layer thickness from the number of laser pulses. It

306   is also evident that the power density has a direct influence on the thickening process, with the

307   larger power density determining a more pronounced thickening for the same number of laser

308   pulses. A maximum total thickness of ~1.25 μm was achieved, with the process being limited by

309   the large mechanical stresses that build up in the diamond cap layer which ultimately cause local

310   mechanical fractures. For this reason, power densities and number of pulses larger than the ones

311   shown in Fig. 7 could not be tested without incurring in structural damage effects. Consistently

312   with what was previously observed, for all of the structures the second annealing step resulted in

313   a complete removal of the buried graphitic layers upon electrochemical etching. It is also worth

314   stressing that, as explained above, the thickness of the gaps obtained after selective

315   electrochemical etching is systematically smaller than the thickness of the laser-induced

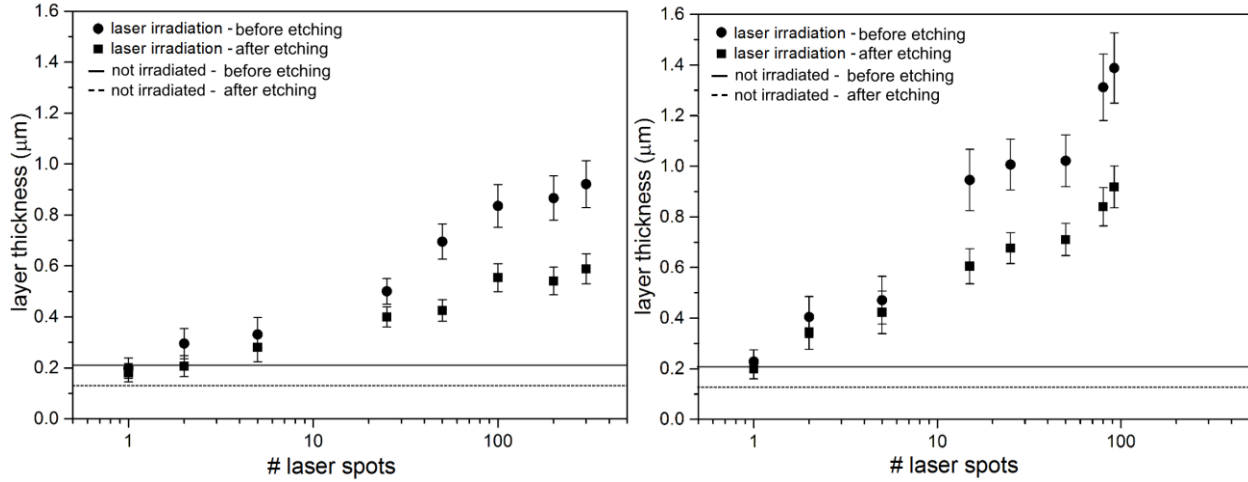316   graphitic layers, as shown in Fig 7.

Fig. 7: Plot of the measured thickness of the sub-superficial graphitic layers versus the number of $\lambda = 532$ nm laser pulses for two different power densities, i.e. a) $P = 0.41$ GW cm^{-2} and b) $P = 0.45$ GW cm^{-2}. The thickness measurements are reported both before (circular dots) and after (square dots) the selective etching process. Likewise, the horizontal lines indicate the thicknesses of the pristine ion-induced layer, both before (continuous line) and after (dashed line) the selective etching process.

4. Conclusions

We reported on the effect of ns-pulsed $\lambda = 532$ nm laser irradiation on the thickening of sub-superficial graphitic layers in diamond obtained by means of 2 MeV He$^+$ ion implantation and subsequent high-temperature annealing. Cross-sectional TEM and EELS measurements elucidated the complex structuring of the processed regions into amorphous and nanocrystalline graphitic multi-layers. The complete conversion of the laser-induced layers to an etchable graphitic phase was obtained only upon thermal annealing at 900 °C, while a highly-oriented phase was found in the residual graphitic pockets after selective etching of the non-annealed samples. A systematic SEM investigation of the thickening of the graphitic region was carried out as a function of the laser irradiation parameters (power density, number of pulses). An

334   increase up to 650% of the initial layer thickness was reached without incurring into critical

335   mechanical failures due to induced mechanical stresses.

336   By allowing a fine tuning of geometrical and structural properties of graphitic layers formed by

337   ion irradiation, laser-induced graphitization offers interesting opportunities for a new level of

338   control in the fabrication of buried graphitic structures in diamond, with appealing applications

339   in different fields in which MeV ion beam lithography and laser graphitization were successfully

340   employed [26,31–36,38,39]

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